

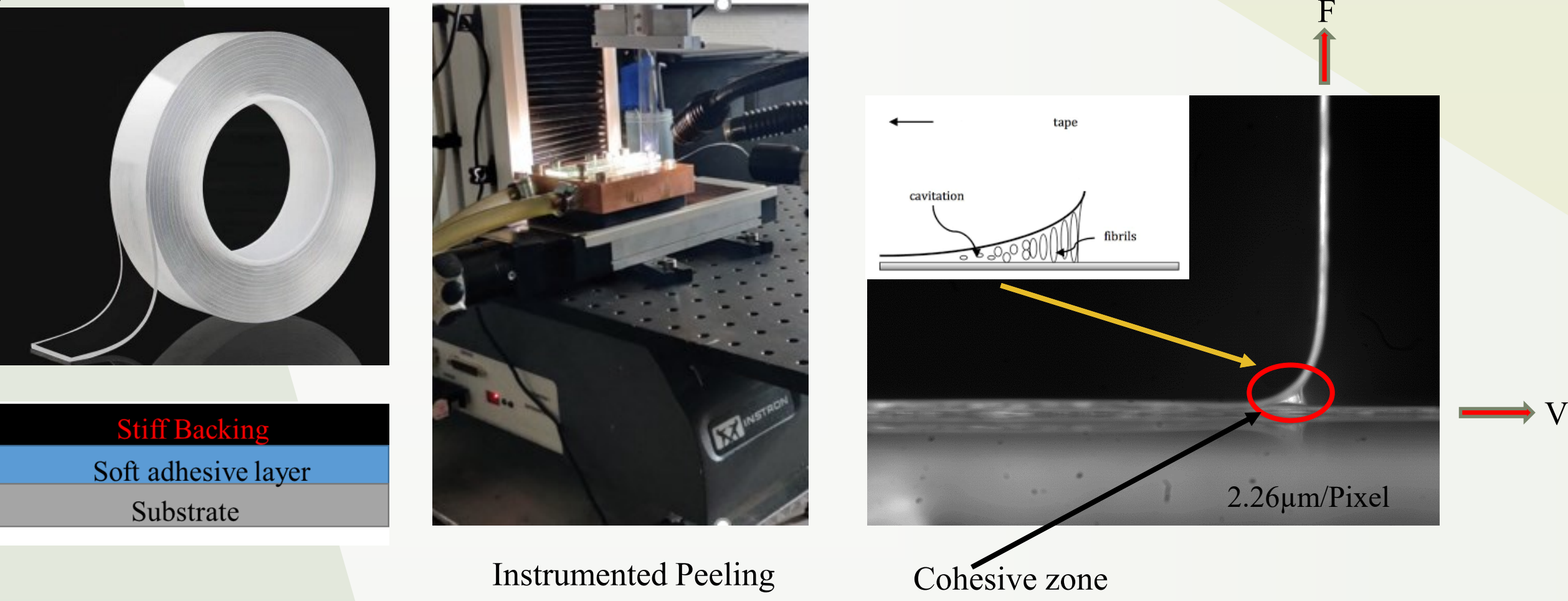
Numerical Modeling of Large Stretch of Adhesive Fibrils

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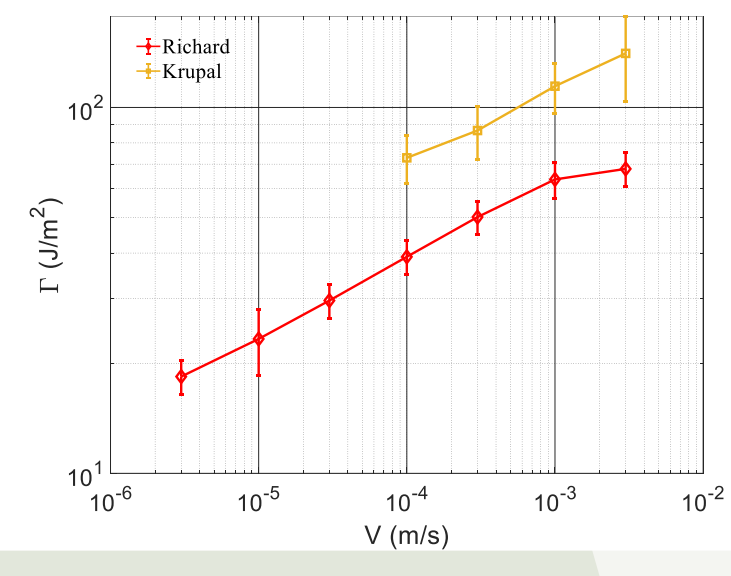


Scotch tape and peeling



Instrumented Peeling

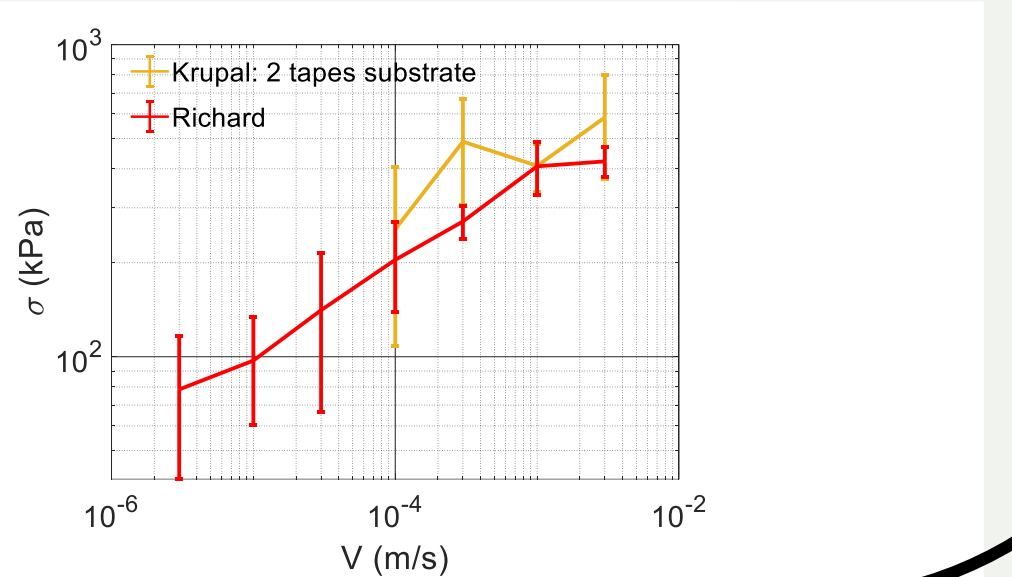
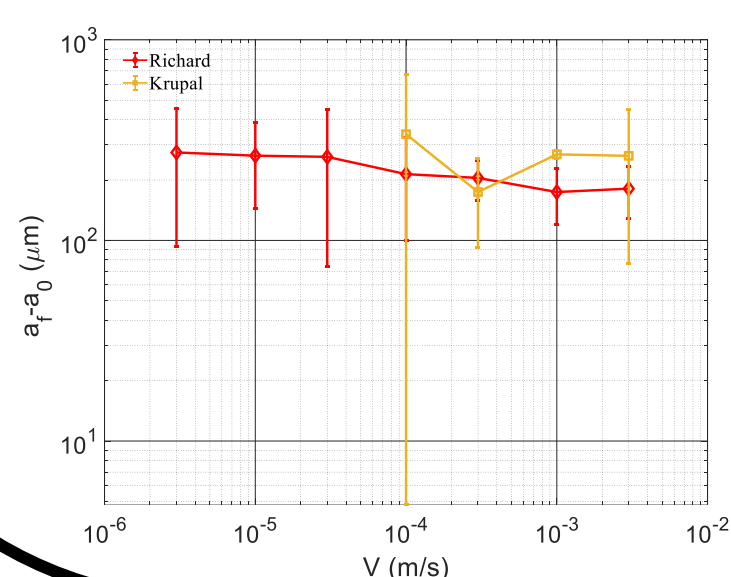
Cohesive zone



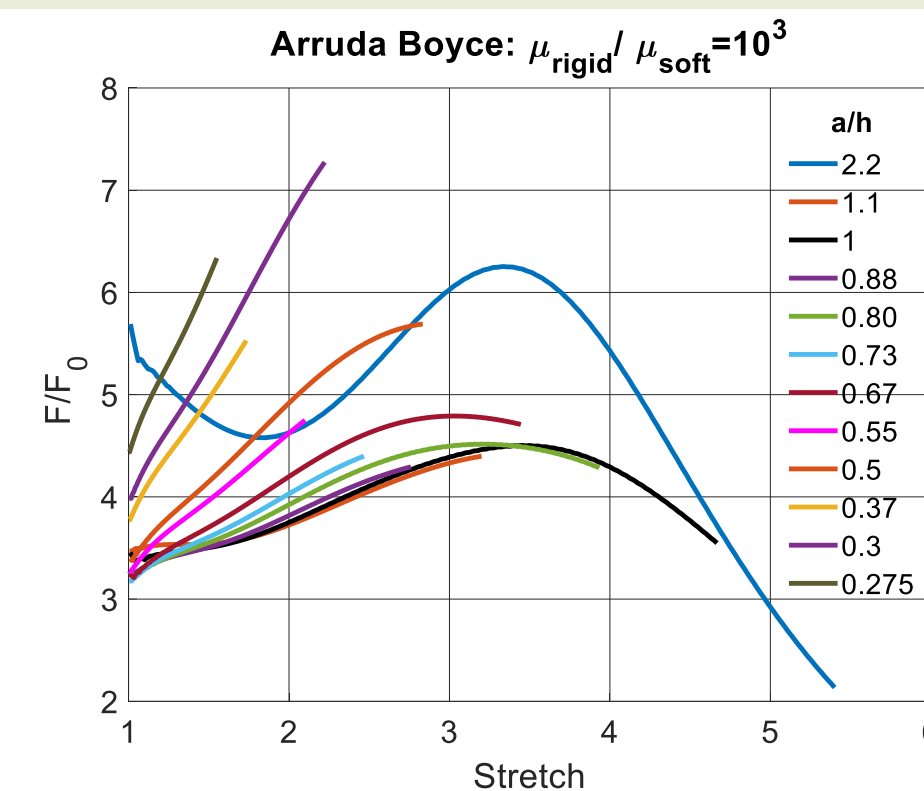
$$\Gamma = \frac{F}{b}$$

b = width of the scotch tape
F = force
Γ = adherence energy
V = velocity

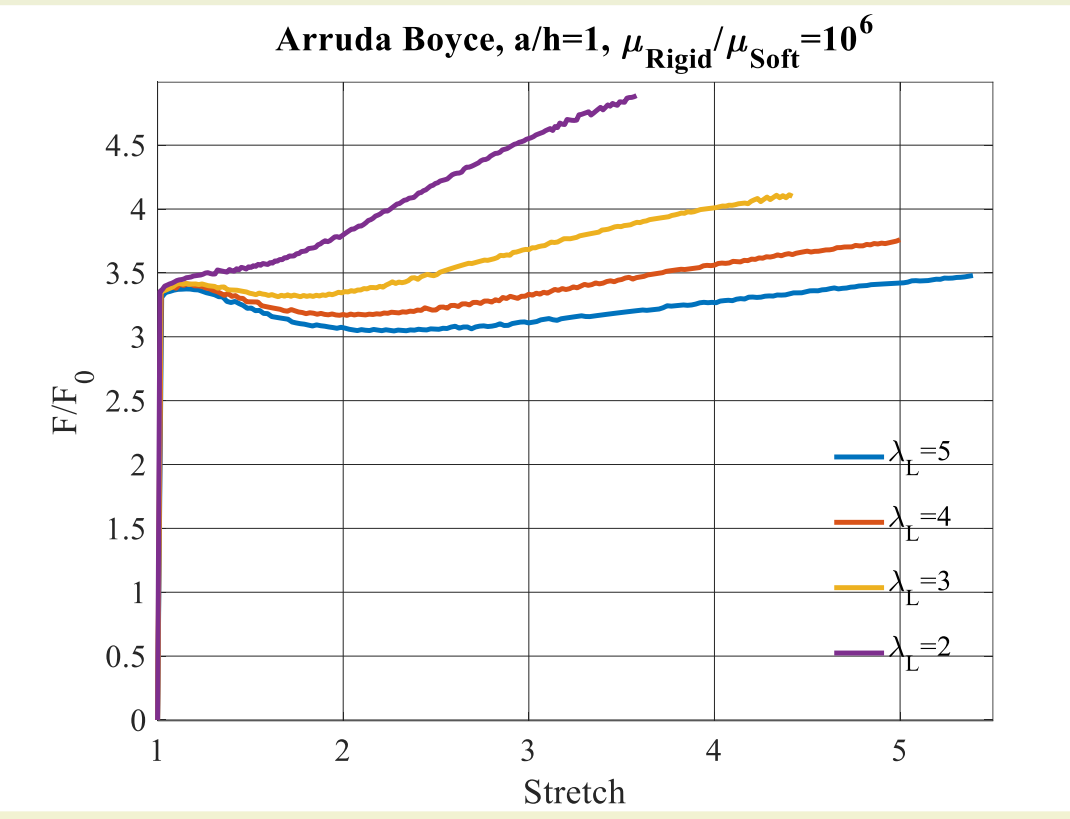
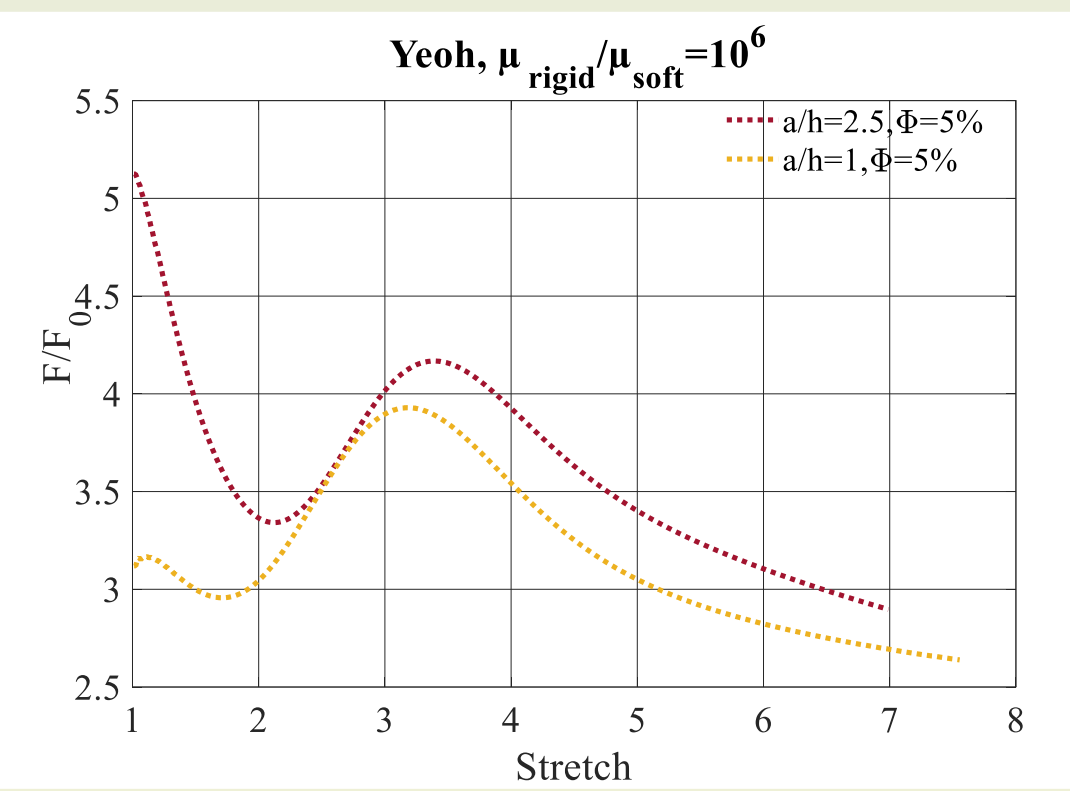
$$\Gamma = a_0 \times \sigma \times \frac{(a_f - a_0)}{a_0}$$



$$\Gamma^c(V) = a_0 k_{exp} \int_0^{\epsilon_f(\epsilon_a)} \sigma(\epsilon, \epsilon_a) d\epsilon$$



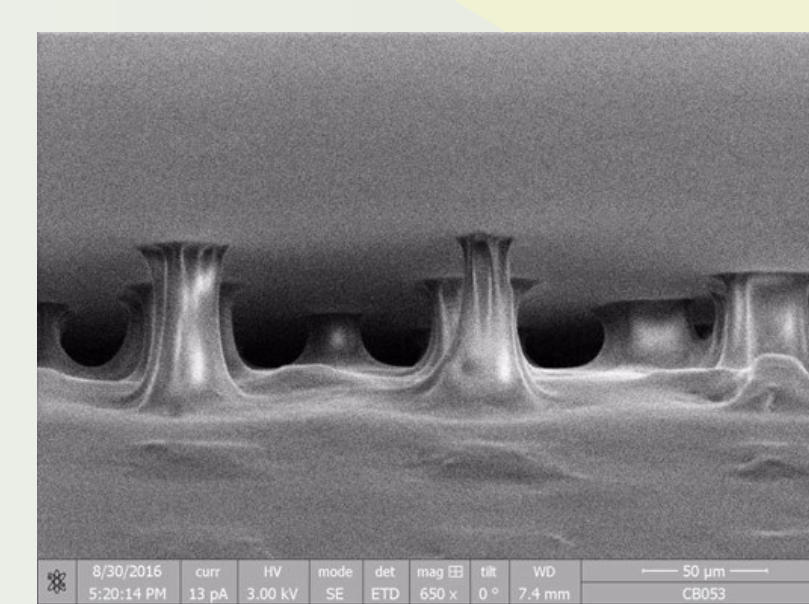
Where, F = force in fibril simulation and F₀ = force in uniaxial simulation for the same stretch, a = radius of the fibril and h = thickness of the soft layer



Simulated Fibril vs Experimental Fibril



Peeling front from bottom on homogeneous substrate



SEM view of adhesion rupture of a PSA tape, showing fibrillation (X. Morelle & B. Bresson)

Yeoh model

a/h=2.5

Φ = 5%

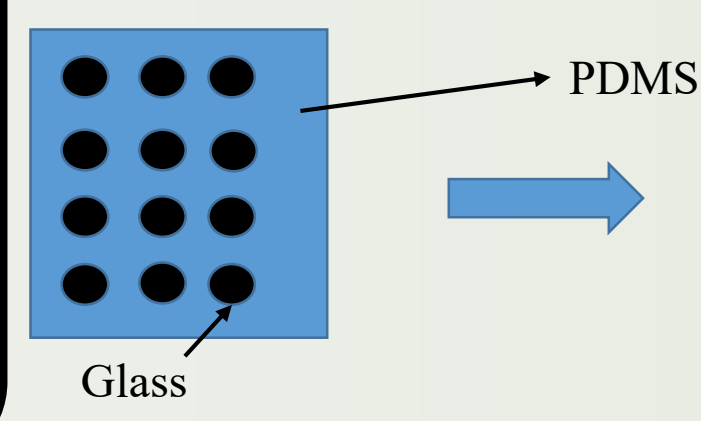
μ = 55kPa

Stretch λ = 4

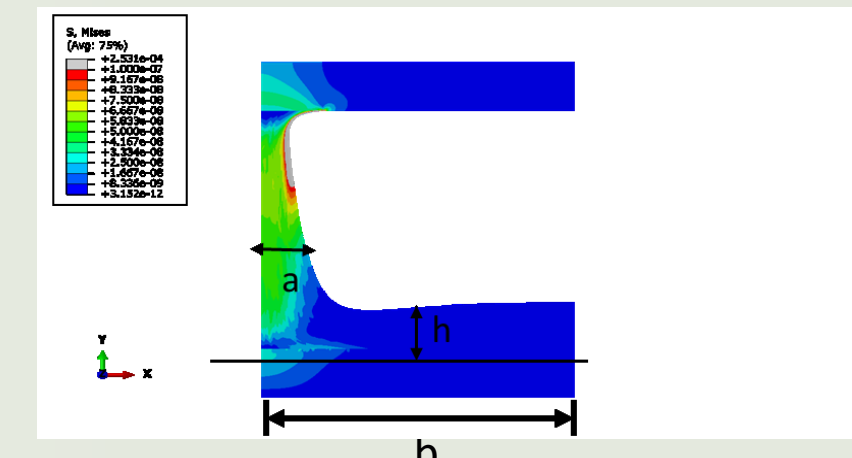
λ = 3.3 s⁻¹

Peeling velocity V = 0.1mm/s

Polymer 6B

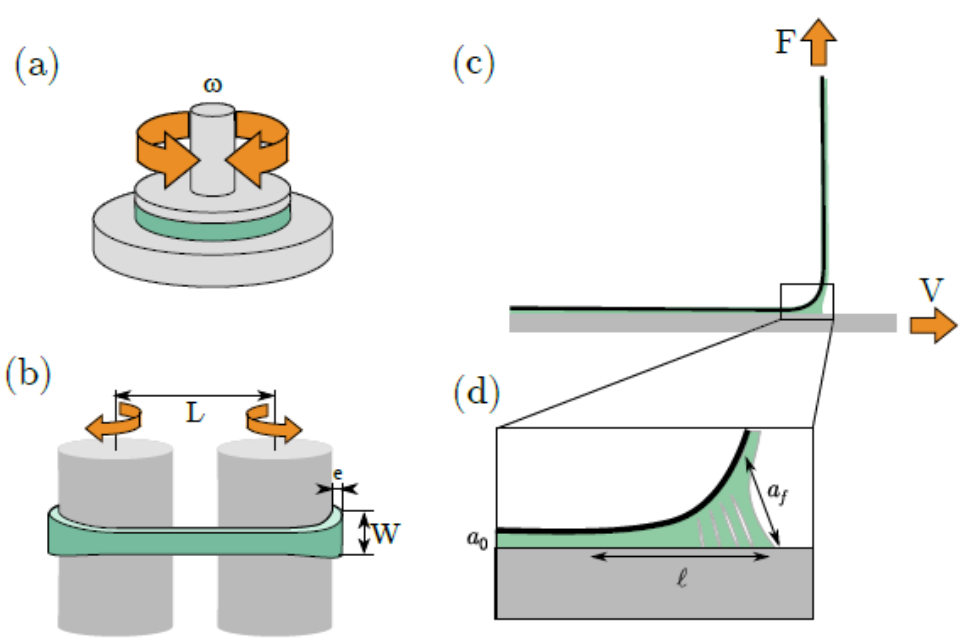


E_{fib} = 2μJ/fibril

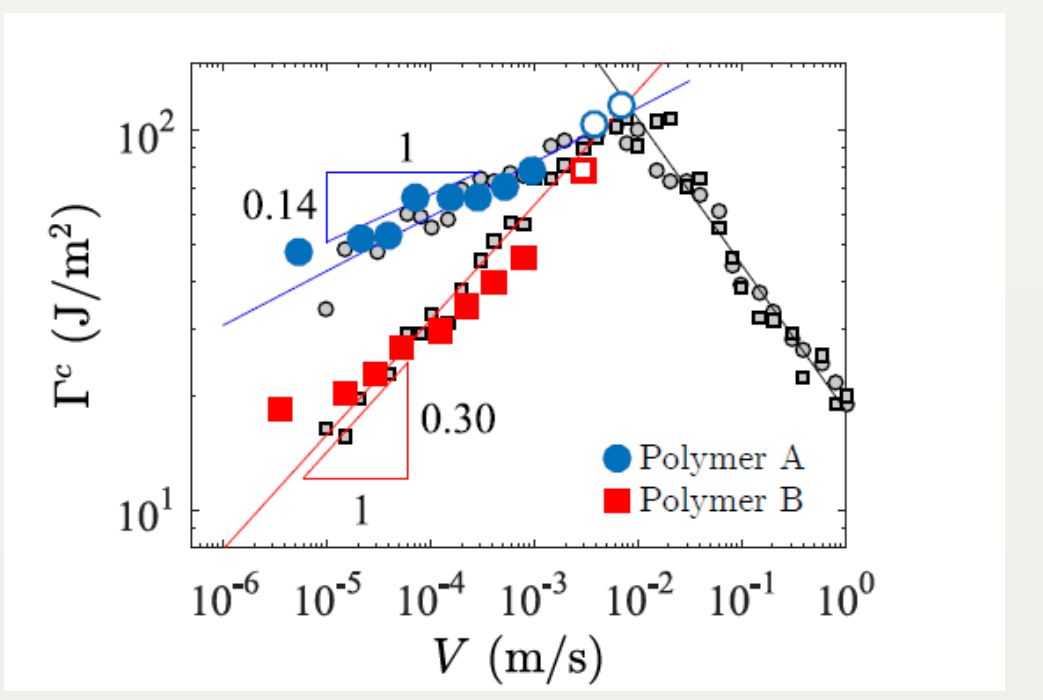


E_{fib} = 0.168 μJ/fibril

Julien's model for peeling



Schematics of experimental setups

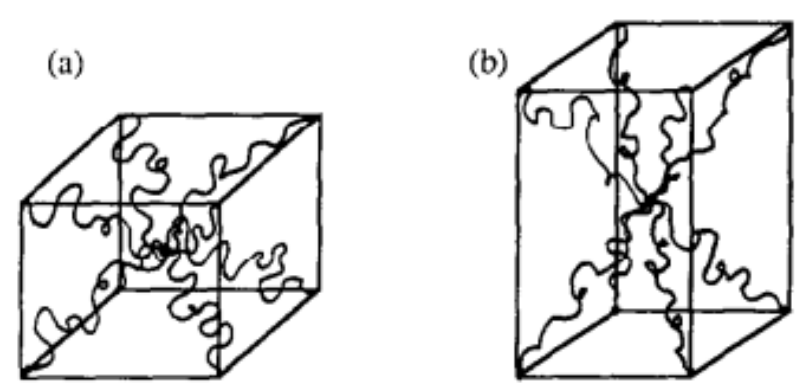
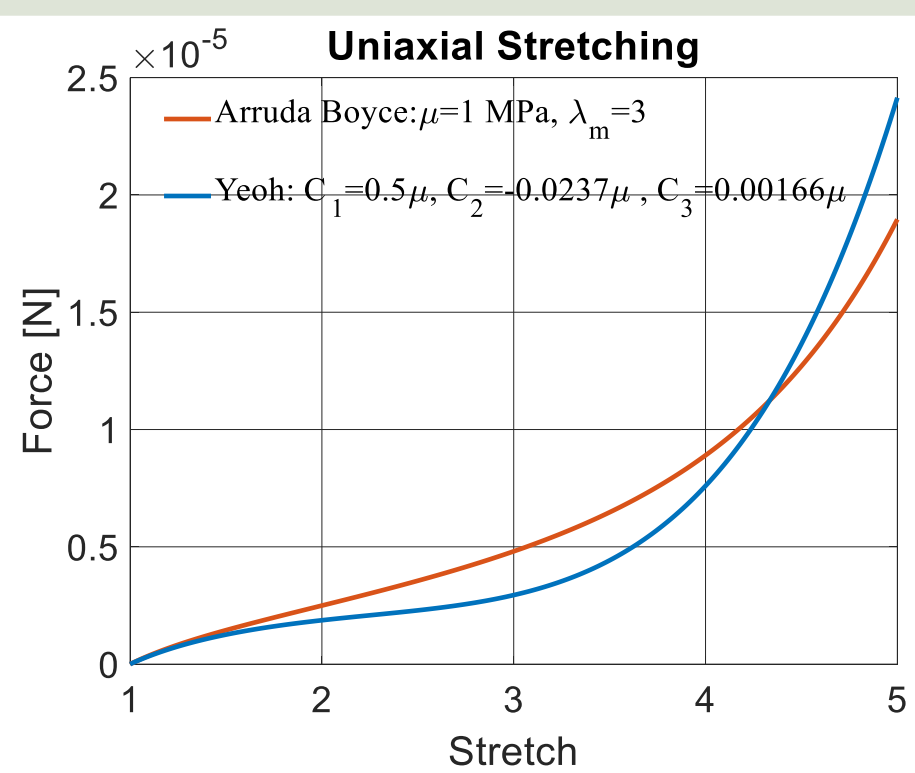


$$\text{Rupture energy } \Gamma^c(V) = a_0 k_{exp} \int_0^{\epsilon_f(\epsilon_a)} \sigma(\epsilon, \epsilon_a) d\epsilon$$

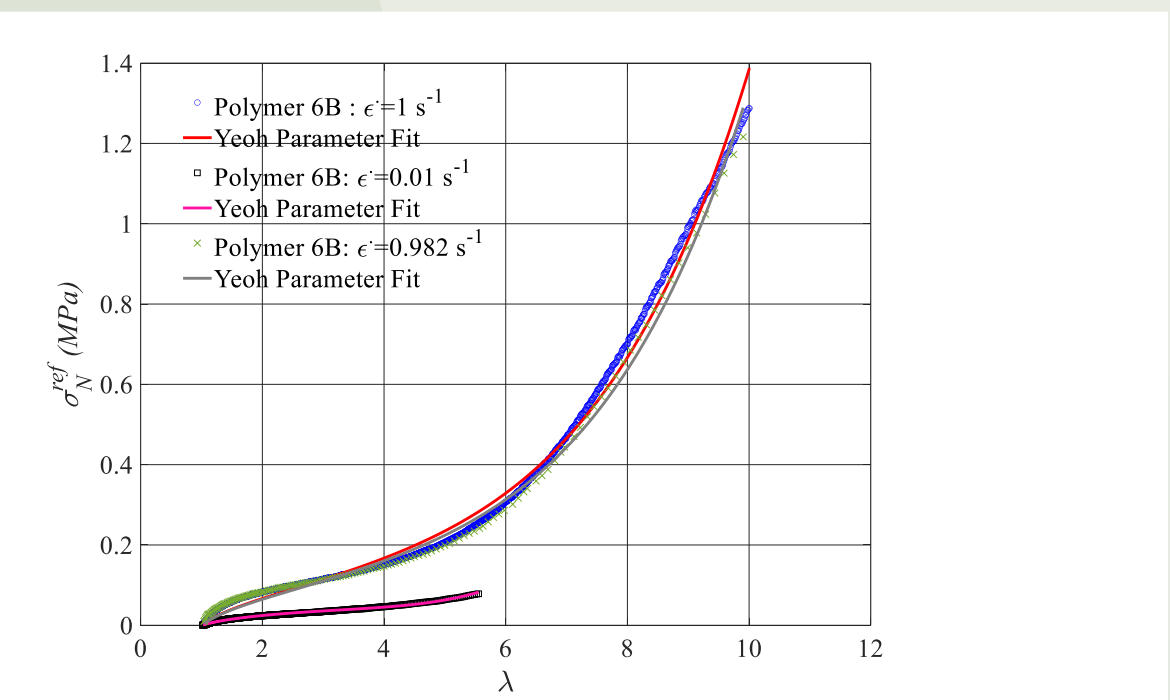
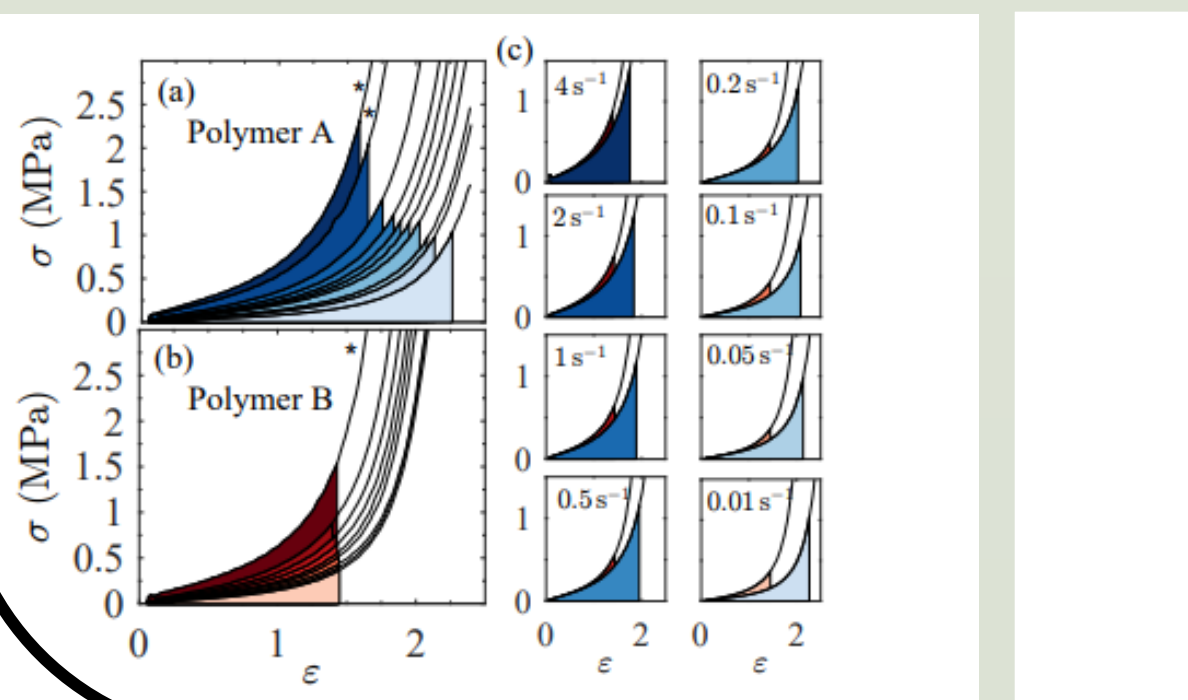
Where l = lateral size and a_f = maximal fibril size
For this geometry energy release rate $G = \frac{F}{b}$
The adhesive layer thickness a₀ = 19 μm

- The dimensionless prefactor k_{exp} can be interpreted as the sign that the fibril drawing process from the bulk adhesive layer is affected by a higher stress triaxiality.
- Value of prefactor: k_{exp} = 5.

Constitutive Relations



Eight chain model for (a) undeformed and (b) uniaxial extension configuration



Validation: Small strain calculations

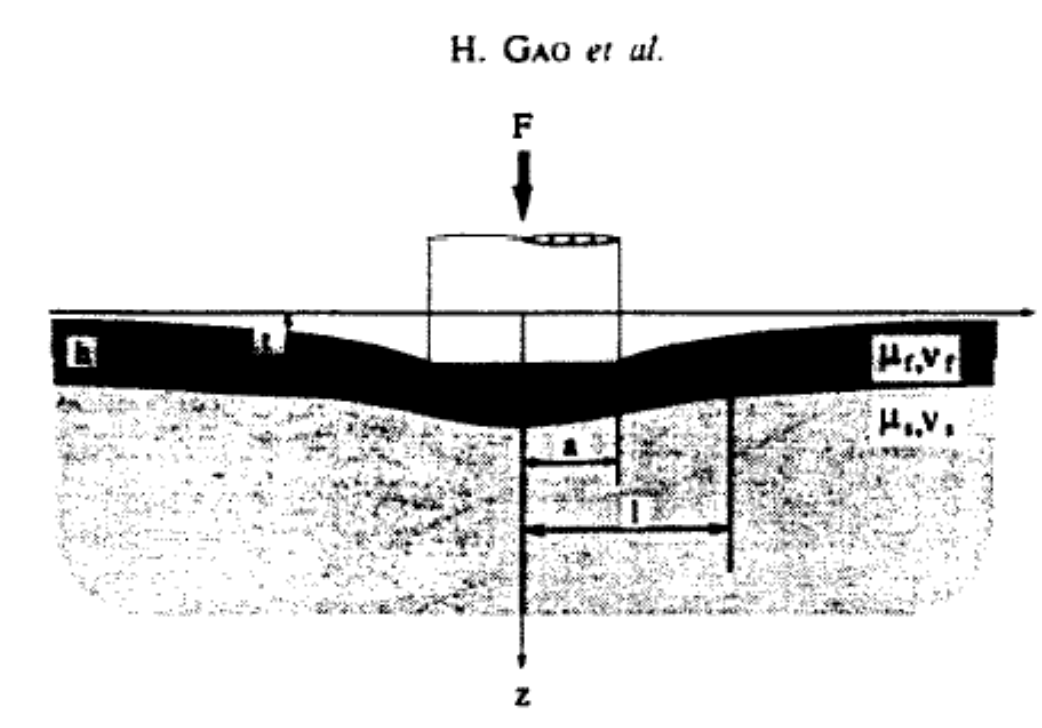
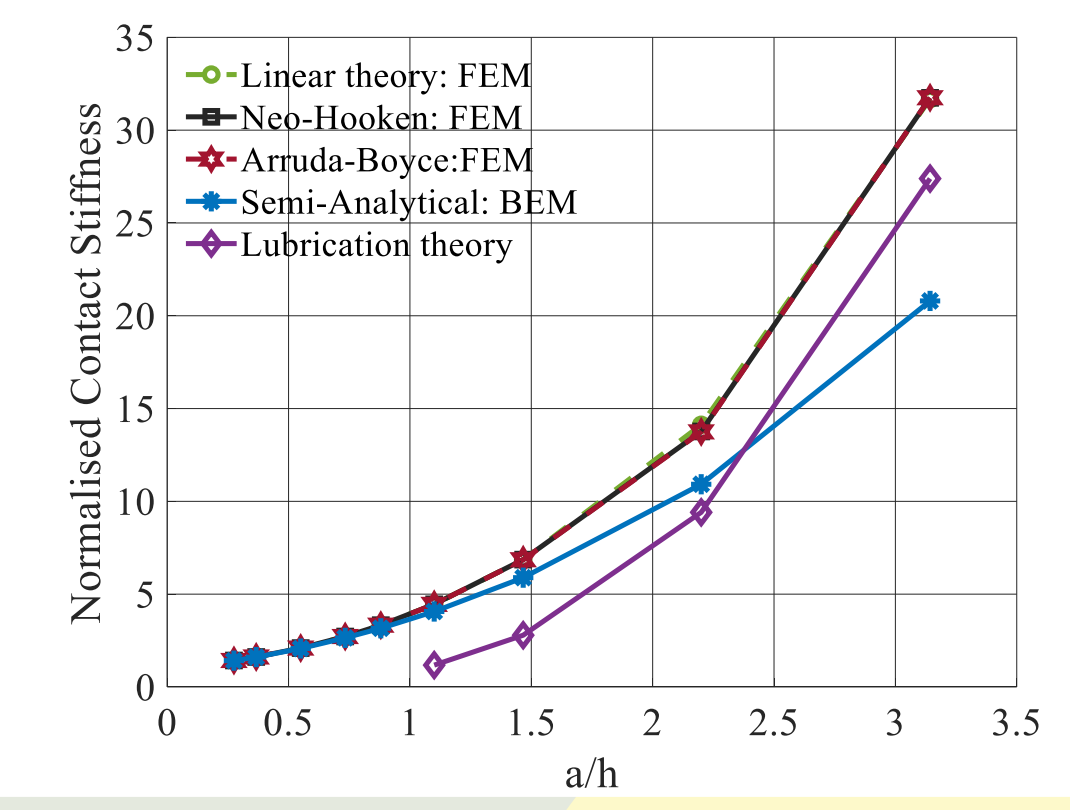
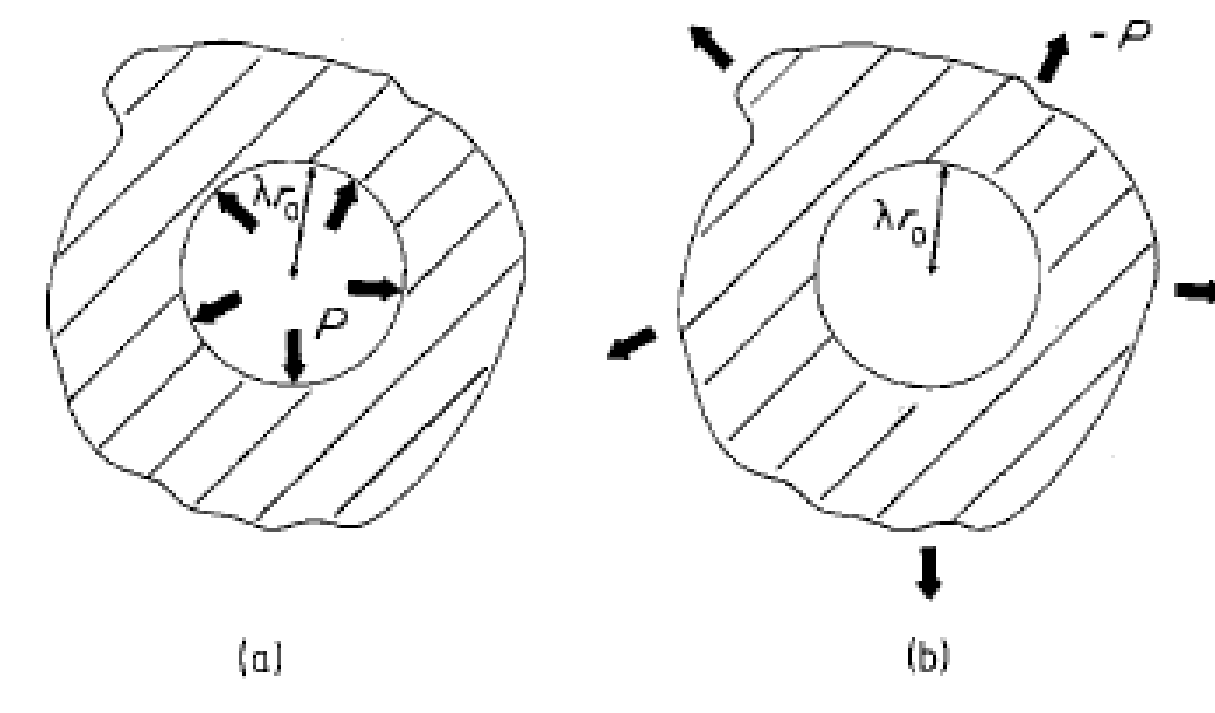


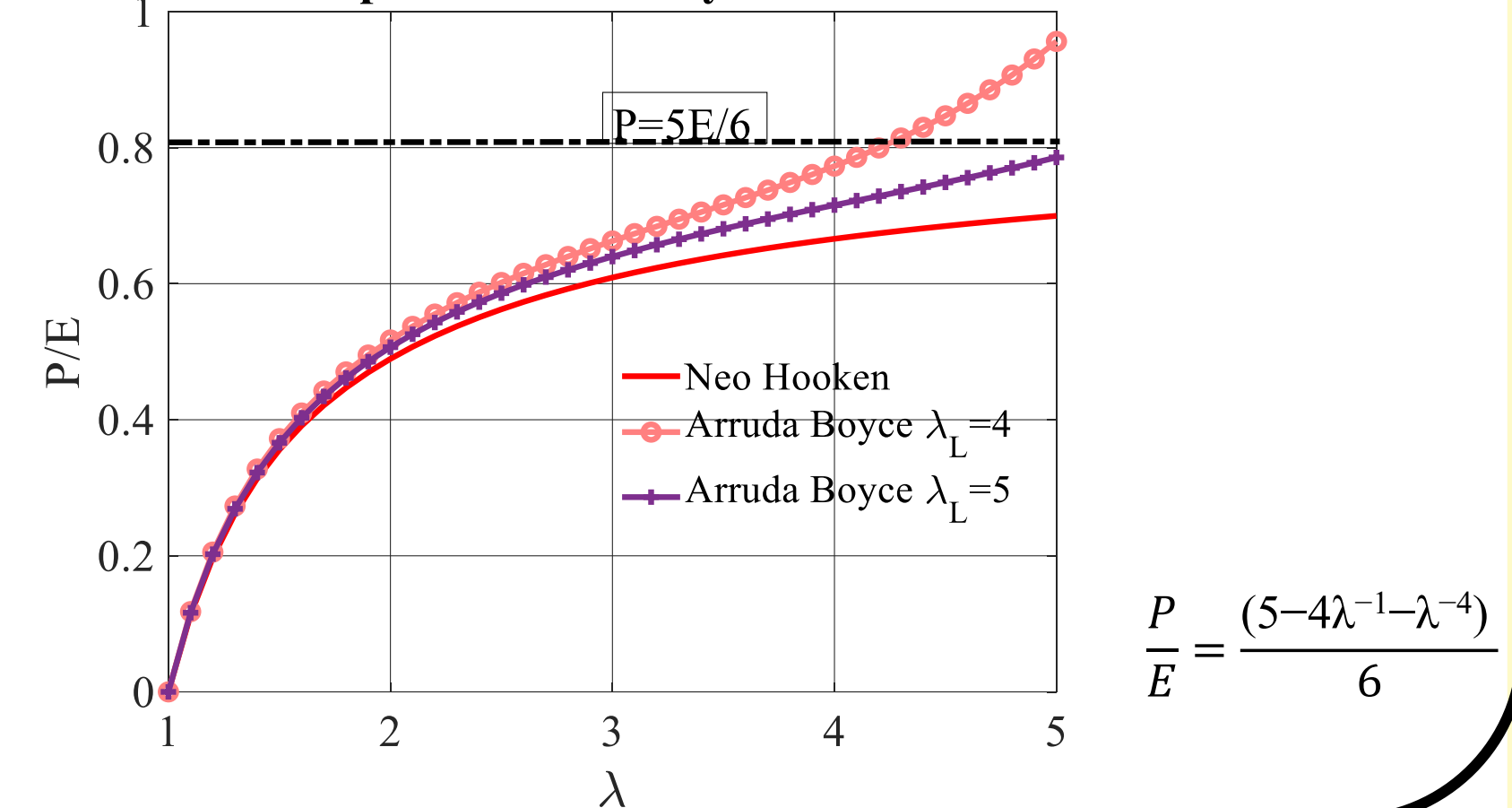
Figure 1 (a) Sketch of an inflated spherical void; (b) spherical void in a medium under a far-field triaxial tension (negative pressure) of -P.



Cavitation

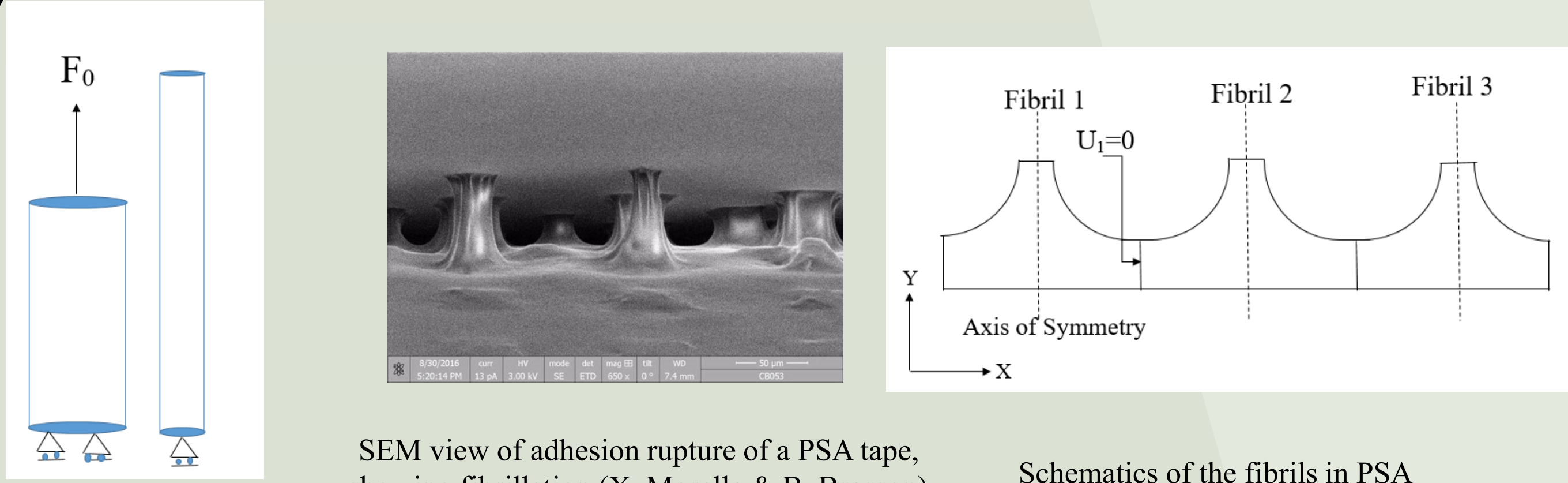


Biaxial expansion of cavity in infinite solid



$$\frac{P}{E} = \frac{(5-4\lambda^{-1}-\lambda^{-4})}{6}$$

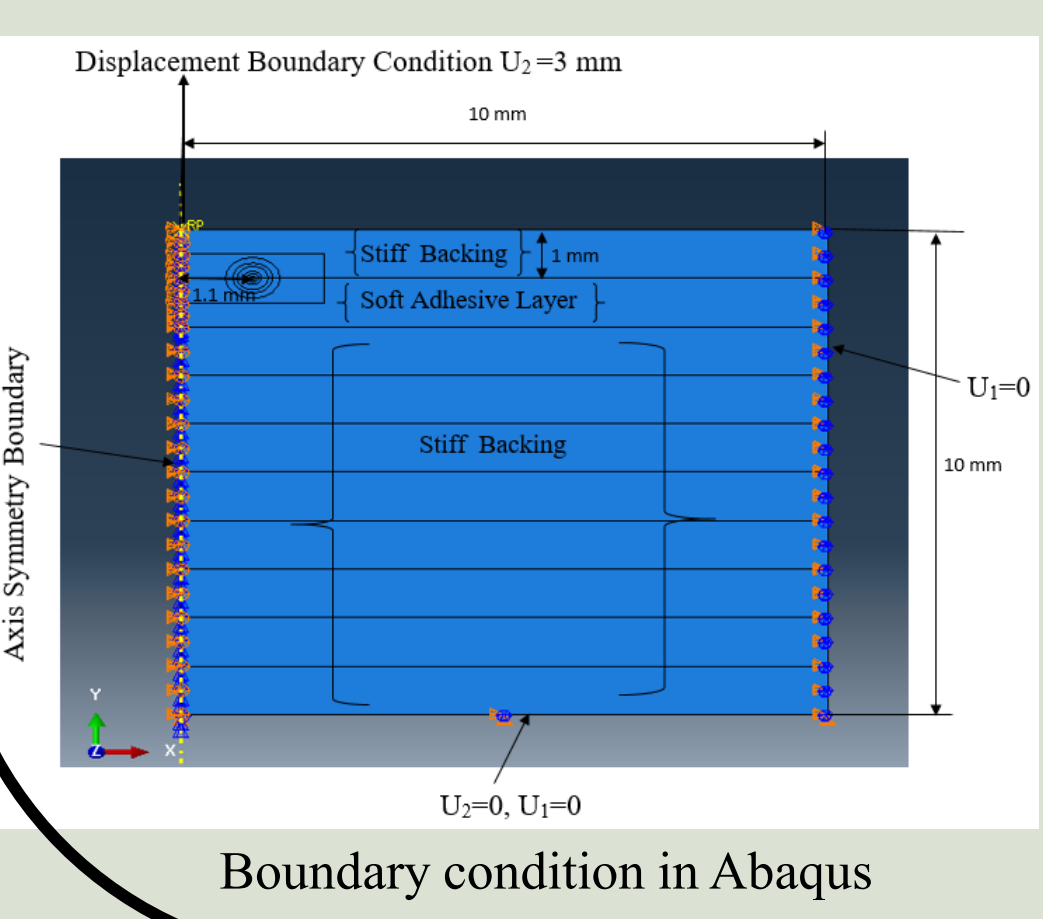
FEM Implementation of modeling single fibril



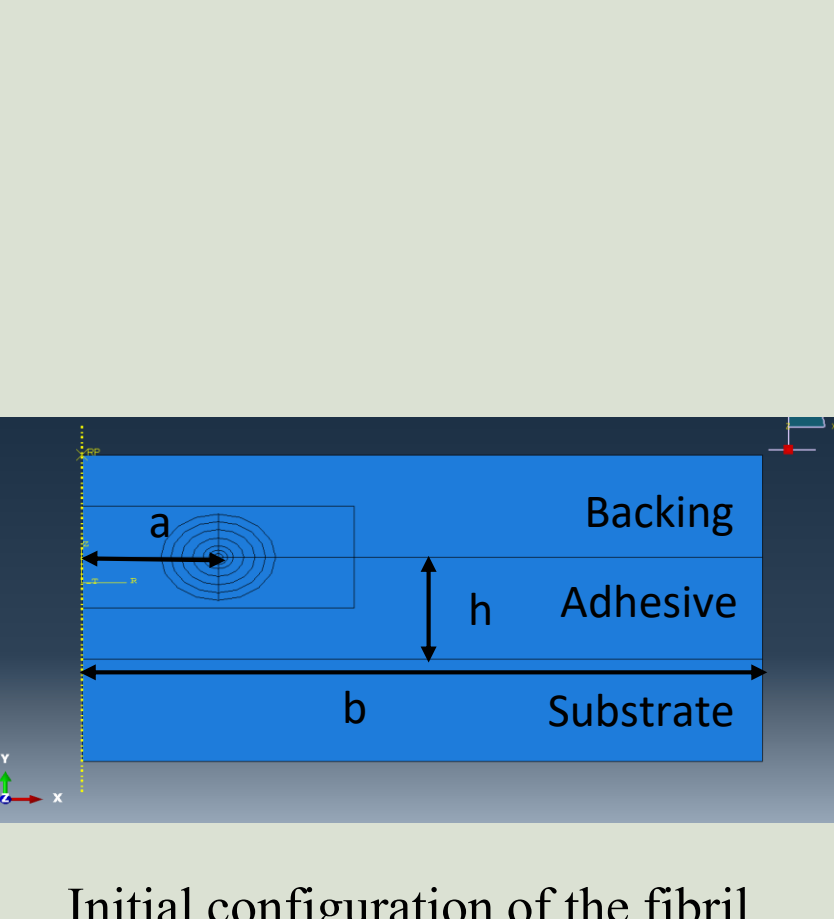
SEM view of adhesion rupture of a PSA tape, showing fibrillation (X. Morelle & B. Bresson)

Schematics of the fibrils in PSA

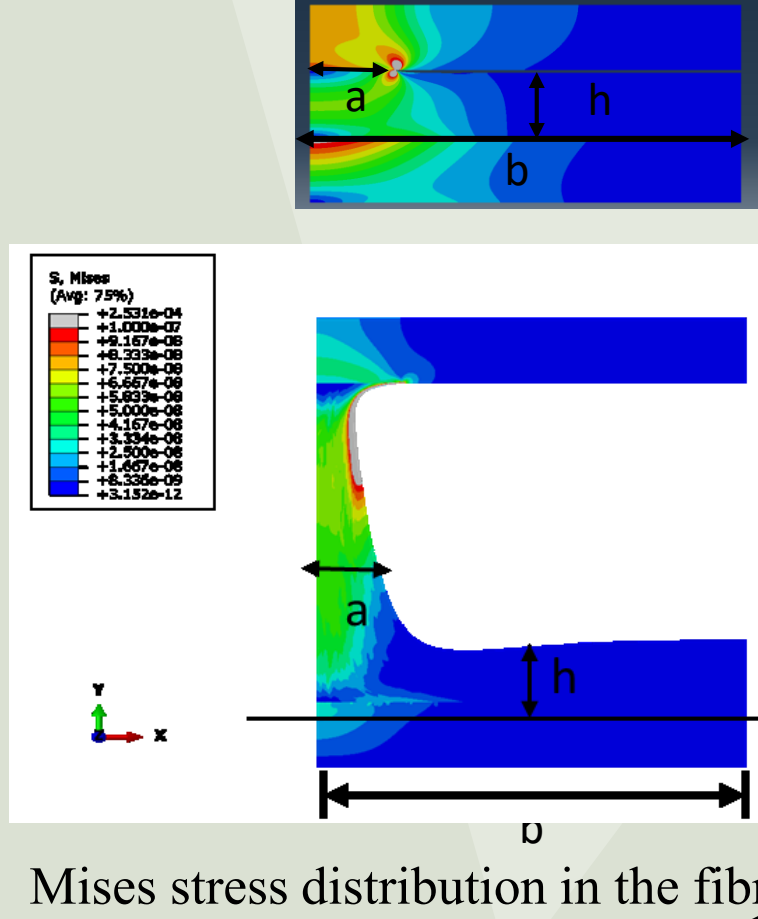
Uniaxial Extension



Boundary condition in Abaqus



Initial configuration of the fibril



Mises stress distribution in the fibril

Conclusions

- At small strain, fibril simulation predicts the factor of 3.3 (k_{simulation} = F/F₀) for a/h = 1.1,1
- This factor comes from the confinement between the backing and the substrate
- Small strain validation of the fibril simulation is in good agreement with the available semi analytical results for moduli mismatch ratio of 100
- Matching simulated fibril energy with experimental debonding energy on patterned substrate require accounting for PDMS contribution
- However, cavitation can be included in the simulation to check whether initial configuration has any role to play in the over all results.
- Cavities in the infinite solid expand without any limit at critical pressure for Neo-Hookean solid. But, if we consider strain hardening and use Arruda-Boyce model than this is not true.

References

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